

# Silicon nanowires: Innovative control growth enabling advanced applications

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Khayyat M.M. Silicon nanowires: Innovative control growth enabling advanced applications. *J Mater Eng Appl.* 2022; 6(2):1-19.

## ABSTRACT

This article reviews the growth concept of silicon nanowires with an attention to semiconductor nanowires filling the gap in the knowledge from the very original work to the very recent innovative experimental work. The objectives of this article are as follows; 1- to describe the original work of epitaxial growth of semiconductor nanowires, 2- to discuss the recently emerged technique of nanoscale templating controlling the growth position of nanowires, and 3- to explore the possible technological applications of position-controlled silicon nanowires. Comprehensive description of the first reported successful Vapor-Liquid-Solid (VLS) 1-D growth of silicon crystals is given. The growth approach of bottom-up and the

supersaturation in a three-phase system of VLS is presented along with the nucleation at the Chemical Vapor Deposition (CVD) processes. Positional assembly of silicon nanowires using current available techniques along with the recently invented one of Nanoscale Chemical Templating (NCT). Several applied and conceptional methods of developing available energy applications using nanowires are included, such as, photovoltaic (PV) cells, Atomic Force Microscopy (AFM) and Metal Oxide Semiconductor Field Effect Transistor (MOSFET) is explained. The final section of this review showed statistical trends in nanowires and Nano rods scientific studies

**Key Words:** Nanowires; Silicon; Aluminum, CVD; catalyst; PV; AFM; MOSFET

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## INTRODUCTION

The topic of Silicon Nanowires (Si-NWs) is a timely emergent study. Over the last few years Si-NWs have come under intensive research as a result of their potential characteristics and conceivable as key materials in advanced optoelectronic applications [1-3]. The review work on a fast-developing topic is not a trivial objective, and it is even more critical with nanotechnology-related subject matters. This review is focused on experimental work and progress of Silicon Nanowires (Si-NWs) technology for the past decades, with more focus on the last decade work. Si-NWs are particularly important, based on the well-known fact of the technological importance of Si material. Moreover, the particular advantages of Nano morphology of high ratio of the area of the surface to the corresponding volume and their related applications [4]. Any application take place at the outer surface of the material such as chemical reactions or photon absorption, it will obviously speed up at that surface of extremely high area [5].

Indeed, there are potential features of Si-NWs to be integrated with

the available applications, such as, Photovoltaic (PV) [6]. The transistor of Metal Oxide Semiconductor Field Effect would benefit from the advancement of Si-NWs in improving performance, such as the Vertical Transport Field Effect Transistor (VTFET). Moreover, some lights have been shed on integrating Si-NWs in developing Atomic Force Microscopy (AFM), Raman spectroscopy, and as stands alone applications, such as sensors [7]. Simulation studies on the possible integration of nanowires to various de-vice fabrication techniques is of great interest at this stage. Particularly, in building specific device structures and studying the expected I-V performance. It has been demonstrated, as an example, using a 3DS quantum simulator of Atlas numerical which is built based on non-equilibrium green's function formalism the physical channel contraction upon nanowires integration with Field Effect Transistors (FET) [8].

The fabrication approaches and progress of growing Si-NWs are based mainly on; bottom-up and top-down techniques.

The direct epitaxial growth of Si-NWs from a catalyzing material on a substrate is called bottom-up growth technique.

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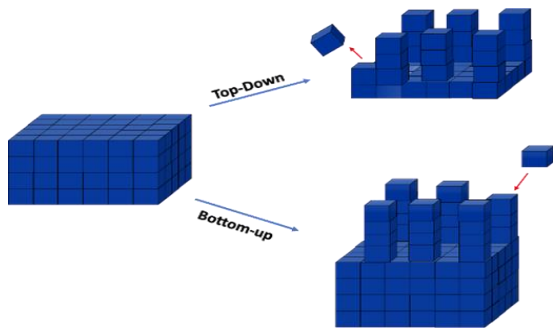
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Received: 01-February-2022, Manuscript No. PULJMEA-22-4178; Editor assigned: 04-February-2022, PreQC No. PULJMEA-22-4178(PQ); Reviewed: 06-February-2022, QC No. PULJMEA-22-4178(Q); Revised: 11-February-2022, Manuscript No. PULJMEA-22-4178(R); Published: 15-February-2022; DOI: 10.37532/puljmea.2022.6(2).1-10



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**Figure 1)** Schematic illustration shows a substrate and the bottom-up process where atoms (the building units) are moving towards or deposited on the substrate, when atoms are moving away or etched from substrate in the top-down mechanism

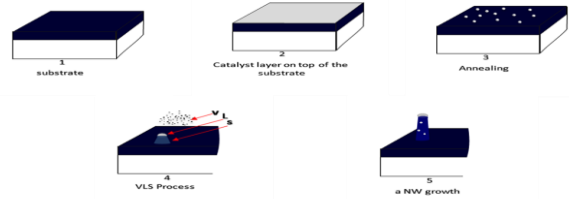
While, photoresist patterning on top of a silicon substrate followed by etching to create vertical structures is explained as top-down approach (see figure 1) [9-12]. Where the details of top-down Si-NWs fabrication approach can be found elsewhere [13,14].

The pioneer work in 1965, carried out by Wagner and Ellis, has led the current work on Si-NWs. The VLS growth system uses metallic droplets or particles (after annealing the sheet layer of metals on the Si substrate), as a seed material nucleating the growth of Si-NWs and adsorb Si gaseous atoms of SiH<sub>4</sub> precursors then precipitate to allow crystal growth. The classical example of growing Si-NWs is the VLS system where gold (Au) metal act as a catalyst eutectic droplet. A recent work by Ramanujan et al has reported the growth and properties of Si-NWs [15-18]. It has been reviewed various growth methods currently employed in bottom-up Si-NWs growth with special attention on Au and non-Au catalysts. Au is the most widely used catalyst for Si-NWs growth by CVD under VLS mechanism, as it offers a good size control. Indeed, there are other growth mechanisms such as vapor-solid-solid (VSS), Solution-liquid-solid (SLS) at the Ultra High Vacuum Chemical Vapor Deposition (UHVCVD) reactor, or using the advanced Molecular Beam Epitaxy (MBE), or laser ablation which have been employed to address issues related to control Si-NWs, such as, diameter, aspect ratio, position. Moreover, catalyst free oxide- assisted methods have also been utilized to grow Si-NWs. Precise positioning of nanowires can be achieved by Electron Beam Lithography (EBL) [19].

Studying the physical properties of the structure of Si-NWs is predominantly critical so that a reproducible relationship between their required functionality and role with their geometrical characteristics can be built. Si-NWs may possess disparate properties due to differences in their crystal phase and directions, crystalline size, i.e., bulk substrates (3D), and nanowires (1D), or thin film or Nano membranes (2D), surface conditions, and aspect ratios. Most studies to- date have used Au as a catalyst for Si-NWs due to the ease of handling that arises from its high resistance to oxidation. Indeed, the interest in other metals to seed the growth of Si-NWs has arisen from the fact that Au impurities in Si decreases the carrier mobility, lifetime, and diffusion length, as Au act as a deep level trap [20]. From the practical point of view, it is desirable to avoid using Au as a catalyst of Si-NWs growth [21]. Based on the previous, the gap in the knowledge of Si-NWs growth and applications is a comprehensive study on Si-NWs catalyzed with elements such as Al which assist the

growth and alloy for advanced applications. The concept of growing semiconductor nanowires is presented in the next session, along with selected resembling tabulated information of growth techniques and catalysts materials. Where semi-conductor nanowires section leads us to a more a specific topic of silicon nanowires and related techniques and applications.

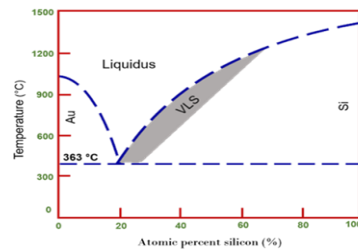
**Epitaxial Growth of Si-NWs**



**Figure 2)** The main five sequence steps of the VLS growth procedure as explained in the text

The bottom-up growth of Si-NWs can be described as shown in figure 2; in the (step 1) depicted in the figure, bulk Si substrate, or a thin grown layer of Si on a cheap substrate such as PC (polycarbonate), PMMA (polymethyl methyl acrylate) or glass. In the (step 2), a few nanometers thick metal catalyst deposited on the surface, which upon annealing it segregates in isolated droplets in the (step 3). Precursor's gas flows in the Ultra High Vacuum Chemical Vapor Deposition (UHVCVD) reactor such as silane (SiH<sub>4</sub>), where Si atoms react at the metal-droplet surfaces and dissolve into solution within the metal (step 4). The catalyst materials supersaturate, inducing precipitation of crystalline Si vapor atoms upon the substrate. As precipitation occurs only at the droplet metal (liquid)-semiconductor (solid) interfaces, the semiconductor atoms crystallite in wire structures with diameters controlled mainly by the diameter of the metal droplet (step 5). This growth practice has been termed by Wagner and Ellis as VLS growth after the three co-existing phases: the vaporous precursors (such as Si<sub>v</sub>), liquid catalyst droplets (such as Au<sub>l</sub>) and the solid silicon substrate (Si<sub>s</sub>). Notice the incorporation, which is quite likely, of some of seed metal atoms within the grown Si-NWs, as it has been described schematically in figure 2.

**Original work of growing Si-NWs: catalysts span from gold to aluminum**



**Figure 3)** Binary-phase diagram for the Au-Si system. The shaded area described by VLS is the range of temperatures at which VLS growth occurs [1].

The cutting-edge technology of growing semiconductors for advanced applications is MBE where the control can be down to atomic level. MBE is an ultra-high vacuum technique that is used when thin films

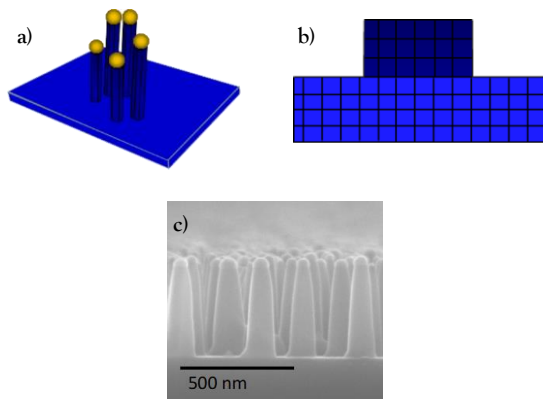
of the highest quality and atomic level perfection are required. Where Shuji Nakamura, awarded the Nobel Prize in 2014 on the invention of the blue InGaN LED using the MBE system. Moreover, the very recent work by Sadeghi et al [36-38]. On growing BaZrS3 chalcogenide perovskite thin films by MBE.

The technique of Nanoscale Chemical Templating (NCT) which was invented in 2013 by Khayyat et al, controlling the position of growing Si-NWs using chemically active catalysts [37]. Based on the binary phase diagram, as shown in Figure 3, of Si and Au, the lowest melting temperature for the Au-Si eutectic is 363°C. The eutectic is lower than the melting points of Au and Si, which are 1064°C, and 1414°C respectively. Considering that the liquid phase of the metal is thermodynamically equilibrated with the solid one of the substrates, the lowering of the melting point, with the size of the droplet is given by equation 1.

$$\delta T = 2\sigma \cdot T_0 / (\rho \cdot L \cdot r) \quad \text{Equation 1}$$

Where  $\delta T$  is the lowering of the melting point,  $\sigma$  is the interfacial-energy,  $T_0$  is the melting point of the bulk metal,  $\rho$  is the material density,  $L$  is the latent heat,  $r$  is the radius of the circle of the catalyst.

As shown in the phase in Figure 3, the eutectic temperature can be summarized as a mixture of two elements at certain proportions that its melting point is much lower than the melting point of either of the two elements that make it up. Thus, annealing the samples which composed of Au film evaporated on Si substrate to the liquid Au-Si eutectic temperature of 363°C. If these Au-Si alloy droplets are placed in an ambient containing a gaseous silicon precursor such as silane ( $\text{SiH}_4$ ), the precursor molecules decompose into Si and  $\text{H}_2$  at the outer surface of the metal droplets, thereby supplying additional Si to the Au-Si alloy.



**Figure 4** (a) Schematic representation of epitaxial growth of Si-NWs, (b) epitaxial growth, i. e; the crystal structure of the grown nanowires is similar to the substrate, (c) SEM micrograph of epitaxial grown Si-NWs on Si(111) substrate catalyzed with Al. [39]

It has been confirmed experimentally that Si-NWs grow perpendicularly on Si (111), as it is represented in Figure 4. However, the growth direction of the Si-NWs by any possible variation on one

or more growth parameters including the growth temperature, which can be attributed in term of surface/interface energy .

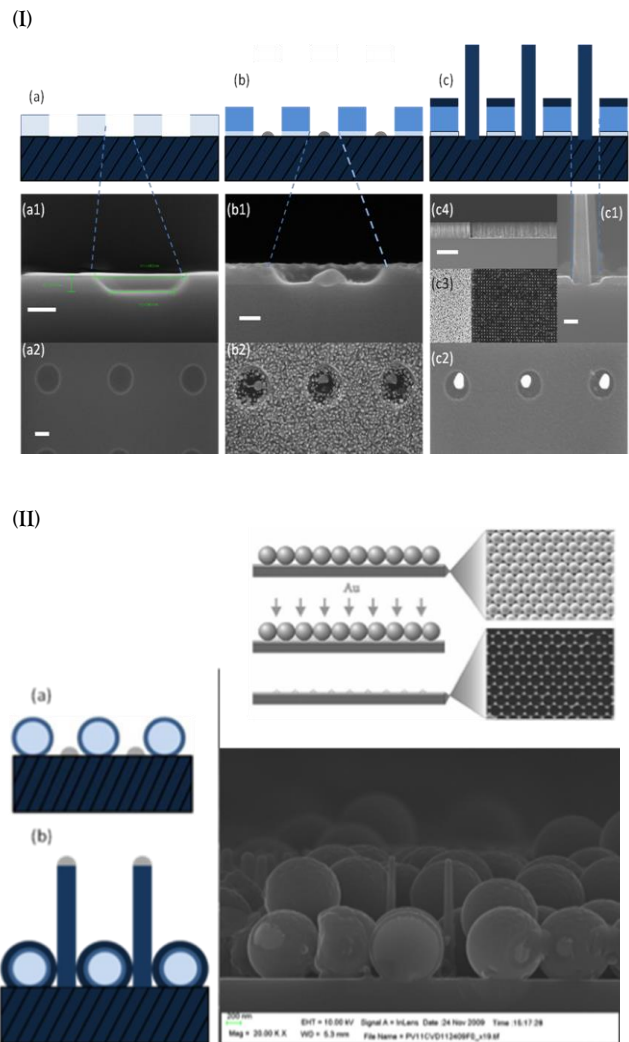
There are a number of CVD systems exist; these can be classified by several parameters mainly the base and operation pressure, such as Ultra High Vacuum Chemical Vapor Deposition (UHVCVD) [25, 26].

In the VLS wire growth the radius of the seed droplet relates to the radius of the nanowire as described in Equation 2.

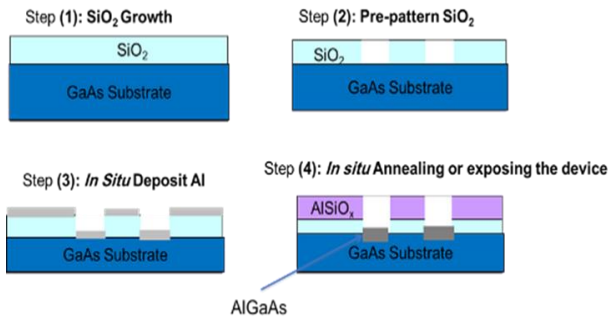
$$R = r\sqrt{1/(1 - (\sigma_{ls}/\sigma_l)^2)} \quad \text{Equation 2}$$

Where  $\sigma_l$  is the surface tension of the liquid catalyst, and  $\sigma_{ls}$  is the surface tension of the liquid catalyst interface,  $r$  is the radius of the Si-NWs, and  $R$  is radius of the seed droplet or catalyst. Studying the various related growth parameters of pressure, temperature and position are of critical importance for implementation of Si-NWs as building units at various applications.

**Innovative approach of growing Si-NWs: Nanoscale Chemical Templating technique**



(III)



**Figure 5** (I) (a)–(c) The NCT technique schematic representation and SEM micrographs. ((a1), (b1), (c1) and (c4)) and plan view ((a2), (b2), (c2) and (c3)). The scale bars are as follows; (a1) and (b1) 100 nm; (c1) 300 nm; (a2), (b2), and (c2) 1  $\mu$ m; (c3) and (c4) 20  $\mu$ m.

(II) Silica microspheres representation of the NCT technique (a & b), and the corresponding SEM micrograph as a proof of the concept. (II-a) The silica microspheres are dispersed on a Si substrate, then Al deposition and the subsequent annealing process, Al reacts with the oxide (SiO<sub>2</sub>) in the microspheres, but the Al droplet on the Si substrate, between the microspheres, are ready to seed the growth of Si-NWs, (II-b). A cross-sectional SEM image, note the bright core of the microspheres, as shown by arrow 1 at both the schematic & SEM of the unoxidized silica (SiO<sub>2</sub>) and the darker contrast (marked as arrow 2) at the microsphere surface where the Al was deposited and the planner growth of Si was occurred.

(III) Selective growth of AlGaAs for further applications, as represented schematically.

It is of special importance to control the spatial placement of Si-NWs for device fabrication. Photo lithography or Electron-Beam Lithography can be used for predefining the precise position of the catalysts, and consequently the spatial placement of nanowires. The topic of controlling the growth position of Si-NWs is an active research topic among several research groups [28-39]. Most of the well-established research projects on positioning Si-NWs for further device integration have used Au.

The innovative approach of the spatial placement of Si-NWs known as Nanoscale Chemical Templating (NCT) has several advanced applications. NCT main application is growing Si-NWs catalyzed Al, which is p-type dopant and a standard metal in silicon process industrial line. The technique is based on patterning a substrate, such as Si, Ge, or GaAs, which is capable of forming alloys with Al during a following annealing step [26].

The concept of the NCT technique arises as a technical innovative solution of the issue of the defective thin planar grown layer (a few nanometers) between the grown Si-NWs, during the time of growing several hundred nanometers of NWs. Now, what does make NCT an innovative solution [26] :

NCT technique is a method involves the following advantages:

- Does not require seed material removal (Al) (see fig. 5 (I)).
- Does not require any lithography steps (see fig. 5 (II)).
- Multiple application space (see fig. 5 (III)).

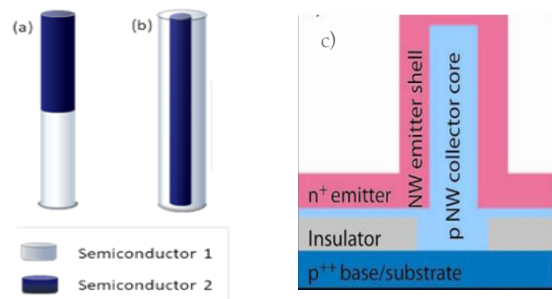
Figure 5 (I) explains how NCT does not require Al removal. (I-a) shows the patterned oxidized surface of the Si substrate forming SiO<sub>2</sub>

layer after photo- or electron beam-lithography. (I-b) shows schematically the sample after Al deposition and annealing, where agglomeration occurs of Al:Si feature in the openings forming the NW seeds, while the Al in contact with SiO<sub>2</sub> has reacted with and roughened the surface. (I-c) shows the NW growth. Notice that a single NW per opening is achieved with fidelity higher than 90%. [38] (I-c3), (I-c4) show a larger area containing both a patterned area and an area with no oxide on the left where random growth appears. Silica microsphere can be used to control the position of the grown Si-NWs, as described in fig.5 (II). In comparison to the previous approach of lithography, here silica microspheres play the role of SiO<sub>2</sub> layer in templating the growth placement of Si-NWs. The schematic representation of spinning silica microsphere, where no lithography is required, on Si substrate, followed by a thin layer evaporation of Al (10 nm) and the sub-sequent step of annealing as shown in (a), where (b) shows the Si-NWs growth between silica microspheres, where growth optimization can be undertaken in future work.

The concept of patterning III-V semiconductor materials selectively is of high industrial importance and it is considered as one of the applications of the NCT technique (III) [26]. This suggested application can be extended to forming novel patterning in III-V semiconductors (see fig. 5 (III)). For example, Al reaction with GaAs will lead to formation of GaAl As selectively in exposed GaAs regions, thereby allowing obtaining patterned GaAlAs and GaAs regions adjacent to each other. Such structures have applications for optoelectronic and FET-like devices.

**Applications on Nanoscale Chemical Templating Technique**

There are several applications of the NCT technique of Si-NWs, such as photovoltaic (PV) made of p-n junctions of NWs as shown in figure 6, or can be used to improve the resolution of Atomic Force Microscopy (AFM) (figure 7), and in Metal Oxide Semi-conductor Field Effect Transistor (MOSFET) to overcome the technological limit of the channel length using Si-NWs meeting the target of miniaturizing as shown in figure 8 [27].



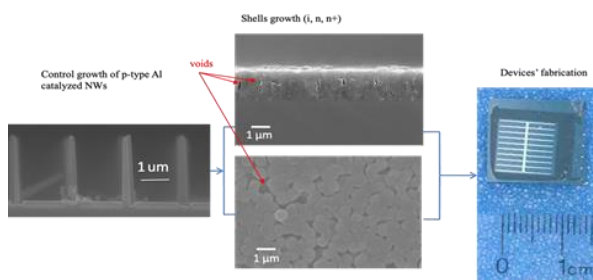
**Figure 6** Schematic representation of (a) Si-NW as an axial heterojunction. (b) Radial heterojunction (c) PV cell as a core-shell [3].

**Photovoltaic Cells**

PV Cells made of Si-NWs p-n junctions have attracted the attention of the scientific community, because of their potential benefits in their short carrier diffusion length across the diameters of the NWs, and their high light absorption. There are several potential benefits of Si-NWs solar cells over conventional bulk Si one or thin-film devices related primarily to cost reduction. This is basically because the Si

substrates do not need to be of high purity to fabricate solar cells of Si-NWs. The potential cost of the PV cells reduces by lowering the purity standard and the amount of Si substrates [6]. There are several research groups are working in developing PV cells based on Si-NWs. The Lieber and Atwater and other groups have developed core-shell growth for their Si-NWs p-n junctions. Moreover, the ability to make single-crystalline Si-NWs on low-cost substrates such as Al foil represents an extra parameter to reduce the overall cost of the cells. Catalyzing the growth of Si-NWS and p or n doping the grown Si-NWs at the same time will be of potential importance for advanced applications. Catalysts can be selected based on phase binary phase diagrams [1, 5, 12, 20, 40].

During devices' characterization it has been noticed that some of the PV Si-NW device are shorted junctions. To study these problem two experiments were designed, to investigate the growth temperature of the capping layers (planner growth without NWs) on Si (100) and Si (111) and measure the shunt resistance vs. position of the grown layer. The shunt resistance varies slightly across the surface of the sample; however, it was low indicating defective planner growth. On the other hand, the growth temperature seems to affect the potential barrier. At this point it was thought that it is important to come up with a method to isolate the planner defective grown areas between NWs from the rest of the device, i.e., Nano patterning. This necessarily eliminates all of the previously described methods of templating the growth, because the metals cannot be protected from oxidation during the patterning processes. The current NCT technique presents a technique that uses the oxygen sensitivity to template the growth of nanowires with Al and other oxygen reactive materials. In NCT technique, SiO<sub>2</sub> layer has been used as a separation layer between the planar defective growths [5, 12]. Rectifying junctions of an array of Si-NWs, catalyzed by Al, were fabricated (see figure 7). The prepared junctions have exhibited slight light sensitivity, which yield relatively low energy conversion efficiency. However, the fact that silicon solar cells based on nanowires have very short p-n junctions which might increases the carrier collection in the core-shell of the nanowire structure.



**Figure 7)** Growing Si-NWs (p-type) doped with Al using NCT, then forming shell of n-type building p-n junctions, where voids appear between formed junctions, finally fabricating the PV cells of 1 cm<sup>2</sup> surface area.

The above stated details of advantages and challenges of PV devices based on Si-NWs can be summarized as follows (Table 1)

**TABLE 1**  
**List of the main advances & challenges of PV cells based**

**on Si-NWs technology.**

<b>Advantages</b>	Short carrier diffusion length, good absorption of light Low cost
<b>Challenges</b>	Si-NWs can be grown on cheap substrates Engineering Depletion Region Width & Density Surface Passivation Isolation between Si-NWs Reducing shunt resistances.

The possible advantages of integrating Si-NWs in relatively large scale solar cells make further investigations worth through simulation and experimental studies for future generation devices. The cumulative effort of various research groups, including ours, have worked to point out the technical challenges which of producing large area (>mm<sup>2</sup>) solar cells from core-shell Si-NWs and other related structures [42]. Rectifying junctions of an array of Si-NWs, catalyzed by Al, were fabricated. The shunt regions between the NWs were identified, and a novel oxidation scheme of NCT technique was employed.

**Atomic force microscopy**

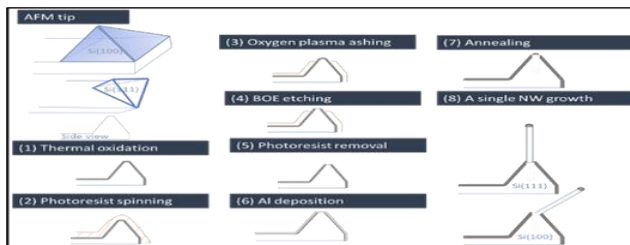
AFM is a machine invented in 1968, for imaging the surface of samples at scales ranging from microns to nanometer, by means of mechanical forces. The AFM consists mainly of three parts: the optical head, the scanner, and the base. The optical head is the main part of the AFM, which is called sometimes the optical sensing system. It is made of a very sharp tip (few nm wide) which is extended down from the end of a small cantilever of SiN or Si (~100 μm long), and an optical system to sense the cantilever deflection.

The fast progress of nanoscience has been benefited from the invention of the AFM, and this development has been increased by the advancement of AFM based on the progress of Si-NWs growth techniques. It has been proposed to improve the resolution of AFM tips in a production scale [27]. The concept of the technique of improving the resolution of the AFM tips is described in figure 8, along with the various steps of the Si-NW growth on the tip of the available Si (100) or Si (111) AFM tips. Figure 8 is showing an AFM tip comprising a silicon cantilever, an etched silicon pyramid formed near one end of cantilever, and a Si-NW is extending from the apex of pyramid. An oxide coating covers the pyramid surfaces with the exception of an opening from which silicon wire was grown. The wire is typically grown by the VLS method in a CVD reactor, which uses a catalyst to promote the wires growth. Typical catalysts may include metals such as Al, Au, Ti, with Al being the preferred choice. The growth of Si-NWs has two components; the first component is the longitudinal growth, which is the growth promoted by the catalyst, and defines the wire length, and the second component is the radial growth. This component is reason for the tapered shape of wire. The radial growth is usually undesirable and needs to be minimized. Lower growth temperature can reduce the amount of tapering by increase the ratio of the longitudinal growth rate to the radial growth rate. However, the use of a lower growth temperature has some disadvantages among which are introduction of crystal defects and a

lower absolute longitudinal growth rate which render the fabrication process more expensive as throughput is reduced.

The use of an Al catalyst provides a unique advantage for obtaining wires with no tapering, i.e. wires that have a constant cross-sectional shape. The wire comprises of two parts, a core and a shell. When Al is used as a catalyst the core will be doped with Al, form an acceptor level in silicon ( $E_a - E_v = 0.067$  eV) and as a result the core will be p-type doped. The shell which forms due to the radial growth remains mostly undoped. A Si etchant such as tetramethylammonium hydroxide (TMAH) or potassium hydroxide (KOH) can now be used to etch electively the undoped shell with respect to the p-doped core. It is showing the wire after the shell was etched.

The grown Si-NW on Si(100) of the squared-base tip is 45° tilted, while Si-NW grow perpendicularly on the tip of the AFM tip of the triangular base of Si(111) [32-35]. Where further reduction of the average wire diameter to the nanometer scale can be done via oxidation or hydrogen annealing [36-43].

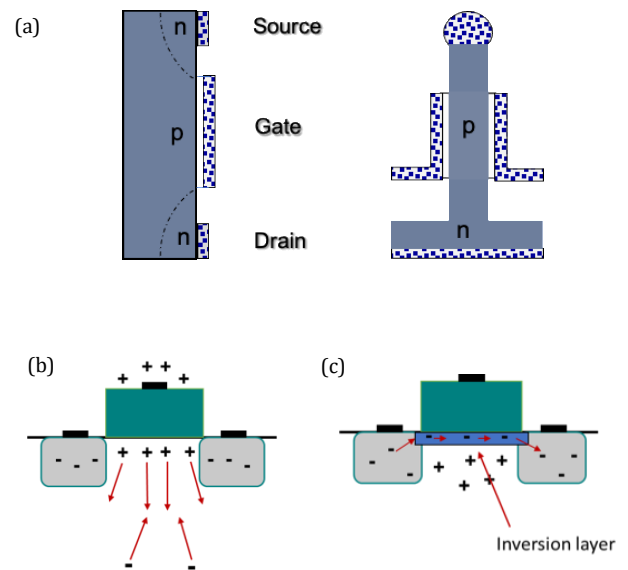


**Figure 8** Schematic illustration of detailed steps of Si-NWs integration with AFM tips [27], the square based Si(100) or the triangle based Si(111) cantilever were first oxidized forming SiO<sub>2</sub> (step 1), spinning photoresist (step 2), oxygen plasma etching the tip (step 3), BOE etching removing the oxide layer at the tip (step 4), photoresist removal (step 5), followed by Al deposition (step 6), annealing to ball up the Al droplet at the tip (step 7), finally Si-NWs growth perpendicularly on Si(111) and tilted on Si(100).

The radius of a Si-NW can be reduced using oxidation sharpening technique. The tip with the grown NW is thermally oxidized at 950 °C for a certain time to oxide the outermost layer the NW, then etch AlSiO<sub>x</sub> in HF. This enables steeply rising steps to be imaged without the result showing the shape of the AFM tip.

AFM can be used to measure surface roughness, scratching, and indentation. The assembled Si-NW scanning tips are suitable for critical topography investigations comparable with the original scanning tips considering the high aspect-ratio nature of NWs and the superior mechanical hardness [44-46]. Moreover, the growth direction can be tailored based on the required AFM investigations. On Si (111), NWs will grow perpendicularly, where as they grow 45° tilted on Si (100) surfaces.

**Metal Oxide Semiconductor Field Effect Transistors**



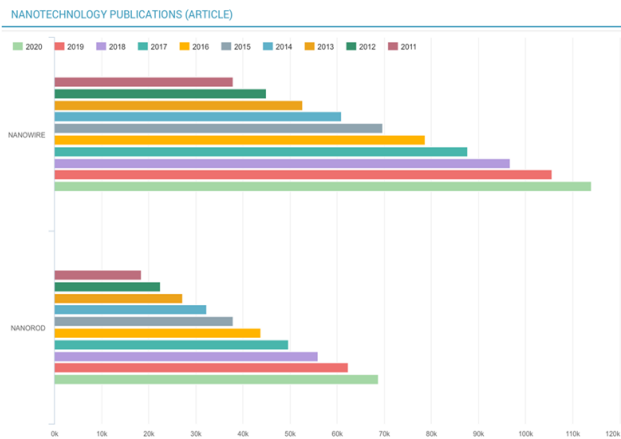
**Figure 9** (a) shows a schematic representation of an (n<sub>p</sub>n) MOSFET the conventional one, in parallel with the innovative one of NWs (b) shows the migrating of charges based on the applied voltages, and (c) presents the formation of the inversion layer, the channel across the diameter of the NW.

MOSFET current technology can be improved in some of its parts using Si-NWs, as shown in figure 9 [8]. Employing Si-NWs as a conduction channel of the npn MOSFET between the source and drain for minimizing the short-channel effects. When a positive voltage is applied to the gate (p-type) the holes in the p-type semiconductor are repelled from the surface (the opposite voltage is correct for pnp MOSFET), then the minority carrier conduction electrons are attracted to the top surface of the transistor. The applied gate voltage exceeds the threshold value, to be followed by an inversion layer formation at the uppermost layer providing a conduction channel between the source and the drain. The width of the conduction channel is controlled by the diameter of the Si-NWs. Where the presented a 3-D schematic diagram of the circular gate-all-around Silicon On Insulator (SOI) nanowire FET with z-axis physical symmetrical contraction, in a structure can be called resonant tunneling nanowire FET (RT-NWFET). They are more key parameters have showed superior properties, where it can be speculated that, the built device via simulation, of RT-NWFET would be an important device for the complementary MOSFET applications.

Moreover, effective integration of Si-NWs and MOSFET will result in modern Complementary Metal-Oxide-Semiconductor (CMOS) technology along with memory applications.

Because of the enhanced surface to volume ratio of NWs, their transport behaviour may be adjusted by altering their surface conditions, and this property may be utilized for sensor applications. Si-NWs sensors will potentially be smaller, operate with less power, and react faster [43-45]. The concept can be extended to forming novel patterning in III-V semiconductors. For example, Al reaction

with GaAs will lead to formation of GaAl As selectively in exposed GaAs regions, thereby allowing obtaining patterned GaAlAs and GaAs regions adjacent to each other. Such structures have applications for optoelectronic and FET-like devices.



**Figure 10)** The total cumulative publications on nanowires & nanorods topics in 10 years (2011-2020, there is no available data yet for the 2021, at the time of writing this article Dec. 2021). Where x-axis represents the number of articles in kilo.

Nanomaterials like the nanowires or nanorods have dimensions within the nanometre range. They are called one-dimensional nanostructures materials. The difference between them and their naming are attributed to the relative ratio between their lengths to their diameters, i.e.; their aspect ratios. The aspect ratio is smaller in case of nanorods; it could be in the range of 3 to 5. While nanowires have diameters of the order of tens of nanometres, with unconstrained length scales, with aspects ratios could be above 1000. The ever-growing demand for smaller electronic devices is prompting the scientific community to produce circuits whose components satisfies size and weight requirements. This demand can be reached by employing Si-NWs, considering their distinct properties, and their combined attributes of cost effectiveness and mature manufacturing infrastructures and use them as nanocomponents to build Nano circuits and Nano biosensors [46-50].

Control of the synthesis and the surface properties of Si-NWs may open new opportunities in the field of silicon Nano electronics. Moreover,

To sum up, semiconductors will continue to inspire us and improving our life quality via continuous dedicative research activities, overcoming the current fabrication barriers [50-54]. MOSFET is the key unit of electronic industries, microprocessors, memory chip, and telecommunications circuits. Based on this, any possible limitations with MOSFET technologies, will consequently affect the other related applications [56, 59]. Moore observed an exponential doubling in the number of transistors in every 18 months through the size reduction of transistor components. This limitation is directly related to the fact that we cannot break down the atomic size barrier, which implies a fundamental size limit at the

atomic/nucleus scale. After all, there is no more direct 18-month doubling, instead there are other forms of transistor doubling may happen at a different slope, which opens doors for more research on nanowires and other Nano technological unit integration. Simulation models of suggested device structures can provide foresight report of the possible approaches of the various available nanostructure integrations [55, 61].

#### ACKNOWLEDGMENT

I would like to thank Dr Davendra K Sadana of T. J. Watson Research Center, IBM, York Town Heights, for his kind support throughout writing this review study. Moreover, I would like to thank my colleagues at KACST for the fruitful discussion.

#### CONFLICTS OF INTEREST

The author declares no conflict of interest.

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